

Table 2. *hkl* reflections and their corresponding *d* and *a* spacings for NiFe<sub>1.95</sub>Mn<sub>0.05</sub>O<sub>4</sub> sintered at 1200°C under 1.013 × 10<sup>5</sup> N m<sup>-2</sup> oxygen

Film shrinkage = 1.0003.

| Trial No. | <i>hkl</i>       | $\theta(^{\circ})$ | <i>d</i> (Å) | <i>a</i> (Å) | <i>a</i> (2 $\theta = 180^{\circ}$ ) (Å) |
|-----------|------------------|--------------------|--------------|--------------|--|
| 1         | $\alpha_2$ 5 5 5 | 68.298             | 0.96268      | 8.33707      | 8.3417 ± 3 × 10 <sup>-4</sup>            |
|           | $\alpha_1$ 6 6 2 | 69.269             | 0.95637      | 8.33748      |  |
|           | $\alpha_2$ 6 6 2 | 69.594             | 0.95642      | 8.33788      |  |
|           | $\alpha_1$ 8 0 0 | 73.613             | 0.93232      | 8.33897      |  |
|           | $\alpha_2$ 8 0 0 | 74.060             | 0.93224      | 8.33824      |  |
|           | $\alpha_1$ 9 1 1 | 77.694             | 0.91549      | 8.34047      |  |
| 2         | $\alpha_2$ 5 5 5 | 68.281             | 0.96280      | 8.33809      | 8.3412 ± 2 × 10 <sup>-4</sup>            |
|           | $\alpha_1$ 6 6 2 | 69.254             | 0.95647      | 8.33830      |  |
|           | $\alpha_2$ 6 6 2 | 69.581             | 0.95650      | 8.33856      |  |
|           | $\alpha_1$ 8 0 0 | 73.605             | 0.93236      | 8.33929      |  |
|           | $\alpha_2$ 8 0 0 | 74.070             | 0.93220      | 8.33783      |  |
|           | $\alpha_1$ 9 1 1 | 77.691             | 0.91549      | 8.34055      |  |

where  $r_A$ ,  $r_B$  and  $R_O$  stand for the radius of the *A* site, the *B* site and the O ion, respectively, and *a* and *U* have the usual significance. In the case of the sample with  $x = 0.05$  Mn, as well as having a larger *U* parameter, it also has a larger lattice constant *a*. Such a combined effect can arise from an enlargement of the *A* sites of this sample. In the case of the sample with  $x = 0.02$  Mn, the results lead us to suppose that the impurity in the sample may not be wholly incorporated in the lattice. It seems likely that the impurity remains at the grain boundary. Electron-probe investigations might help to clarify studies on the grain boundaries of such a sample.

## References

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**Die Strukturbestimmung der Isomaltulose, C<sub>12</sub>H<sub>22</sub>O<sub>11</sub>·H<sub>2</sub>O: errata.** Von WOLFGANG DREISSIG und PETER LUGER, *Institut für Kristallographie der Freien Universität Berlin, Takustrasse 6, 1 Berlin 33, Bundesrepublik Deutschland*

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The following corrections are made to the data given in the paper by Dreissig & Luger [*Acta Cryst.* (1973), B29, 514–521]. The positional parameters (×10<sup>3</sup>) in Table 3 should read: H(6):  $x = 171$  (5),  $y = 172$  (4); H(9):  $x = 238$  (4). The bond angles in Fig. 3 should read C(10)–C(11)–C(12) 111.5, C(2)–C(3)–C(4) 103.0°.

Alle Daten sind in der Zusammenfassung gegeben.